

## AN ANISOTROPIC ELASTOPLASTIC MODEL BASED ON AN ISOTROPIC FORMULATION

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### ABSTRACT

This paper shows a generalization of the classic isotropic plasticity theory to be applied to orthotropic or anisotropic materials. This approach assumes the existence of a real anisotropic space, and other fictitious isotropic space where a mapped fictitious problem is solved. Both spaces are related by means of a linear transformation using a fourth order transformation tensor that contains all the information concerning the real anisotropic material. The paper describes the basis of the spaces transformation proposed and the expressions of the resulting secant and tangent constitutive equations. Also details of the numerical integration of the constitutive equation are provided. Examples of application showing the good performance of the model for analysis of orthotropic materials and fibre-reinforced composites are given.

KEY WORDS Fictitious isotropic stress Anisotropic elastoplastic model Multiphase materials

### INTRODUCTION

The formulation of adequate constitutive laws for orthotropic or anisotropic solids is a problem of large complexity.

One of the more popular attempts to formulate yield functions adequate for orthotropic materials is due to Hill who succeed to extend the Von Mises isotropic model to the orthotropic case<sup>7</sup>. The main drawback of Hill's theory is its limitation to simulate the mechanical behaviour of geomaterials, wood and composite materials. Different modifications of Hill's orthotropic model, for the analysis of fibre-reinforced components have been reported by different authors<sup>20-23</sup>.

The idea to formulate the behaviour of an anisotropic material by means of an equivalent isotropic solid was first introduced by Betten using the concept of *mapped stress tensor*<sup>1-3</sup>. This concept offers the possibility of using all the advantages of the well known isotropic models; consequently it has many computational advantages.

The authors have developed a generalization of standard isotropic plasticity theory for the analysis of anisotropic solids in previous works<sup>18,25,26</sup>. The basic idea was to model the behaviour of an anisotropic solid by means of a fictitious isotropic solid. A basic assumption of the model was that the elastic strain is unique for both the real and fictitious spaces. This situation introduces a limitation in the anisotropic mapped theory, because it involves a proportionality concept between the yield strength and the elasticity modulus for each material direction ( $f_{11}^S/E_{11} = f_{22}^S/E_{22} = \dots = f_{23}^S/E_{23}$ ). In the present work a generalization of such basic theory is introduced.

The anisotropic behaviour is formulated by means of the *fictitious isotropic stress and strain tensors* which results from the tensor transformations of the real stresses and strains. This allows

us to use the same yield and potential functions derived for standard isotropic materials, whereas all the relevant information on the material anisotropy properties is embedded in the two fourth order transformations tensors only. The material parameters involved in these tensors can be defined from adequate experimental tests. This model is here termed 'isotropic mapped model for non-proportional materials'.

The formulation presented is completely general and it allows us to model different class of orthotropic and anisotropic materials typical in composites. The model seems to be particularly suited to be applied for analysis of multiphase materials such as fibre-reinforced composites and concrete.

In the next section the properties of the constitutive law for a general anisotropic material are described together with the basic ingredients of the stress and strain transformations and the derivation of the second and tangent constitutive relationships.

Details of the integration of the constitutive equation are given next. The implementation aspects into a finite element code are detailed in a later section.

Finally, the model is applied to three particular problems: (1) the study of the fibres orientation in a fibre-reinforced composite; (2) the comparison of the results in the analysis of an orthotropic material with those provided by standard Von Mises–Hill<sup>7</sup> theory; and (3) the analysis of a fibre-reinforced composite material.

## GENERAL CONSTITUTIVE LAW FOR AN ANISOTROPIC PLASTIC MATERIAL

A general anisotropic plastic model based on a consistent thermodynamic approach is presented. The model is formulated in a material configuration using total Lagrangian kinematics<sup>6,12</sup>. The model, as presented here, can deal with non-linear problems involving large plastic strains and small elastic strains<sup>9,12,17</sup>. Obviously, simpler orthotropic and isotropic plasticity models are readily obtained as particular cases of the model proposed.

*Yield and potential functions; space transformation tensor*

It will be assumed that both yield and plastic potential functions are defined in the Piola-Kirchhoff stress space (material configuration), as:

$$\text{Yield function: } F^S(S_{ij}; \alpha_S^m) = 0$$

$$\text{Potential function: } G^S(S_{ij}; \alpha_S^m) = K \quad (1)$$

where  $S_{ij}$  is the second Piola-Kirchhoff stress tensor,  $C_{ij} = F_{ik}F_{jk}$  is the Cauchy–Green tensor,  $F_{ij}$  is the deformation gradient,  $\alpha_S^m$  is a set of  $m$  internal plastic variables, and  $K$  is a constant parameter.

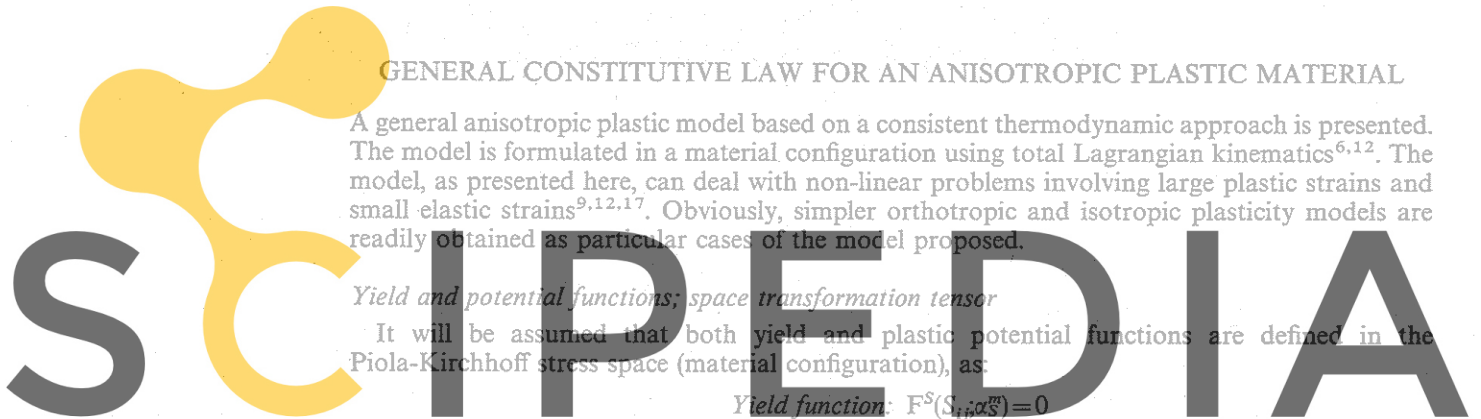
The yield and plastic potential functions are *isotropic* if the invariance condition<sup>1,3,14</sup>:

$$\begin{aligned} F^S(a_{ip}a_{jq}S_{pq}; \alpha_S^m) &= F^S(S_{ij}; \alpha_S^m) = 0 \\ G^S(a_{ip}a_{jq}S_{pq}; \alpha_S^m) &= G^S(S_{ij}; \alpha_S^m) = K \end{aligned} \quad (2)$$

is satisfied for any orthogonal transformation ( $a_{ik}a_{jk} = \delta_{ij}$ , where  $a_{ij}$  is a unit diagonal tensor, and  $\delta_{ij}$  is the Kronecker tensor). Obviously isotropic materials satisfy the invariance condition. This can be simply checked by writing (2) in terms of the first three stress invariant,  $I_1$ ,  $I_2$  and  $I_3$  (see Reference 14), i.e.

$$\begin{aligned} F^S(S_{ij}; \alpha_S^m) &= F^S(I_1(S_{ij}); I_2(S_{ij}); I_3(S_{ij}); \alpha_S^m) = 0 \\ G^S(S_{ij}; \alpha_S^m) &= G^S(I_1(S_{ij}); I_2(S_{ij}); I_3(S_{ij}); \alpha_S^m) = K \end{aligned} \quad (3)$$

For materials satisfying plastic incompressibility (3) can be written in terms of the invariant of



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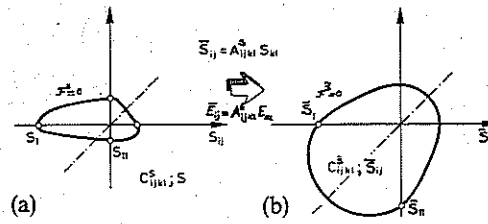


Figure 1 Relationship between the (a) real (anisotropic) and the (b) fictitious (isotropic) space

the deviatoric stress tensor as:

$$F^S(S_{ij}; \alpha^m) = F^S(J_2(S_{ij}); J_3(S_{ij}); \alpha^m) = F^S(I_2(S'_{ij}); I_3(S'_{ij}); \alpha^m) = 0$$

$$G^S(S_{ij}; \alpha^m) = G^S(J_2(S_{ij}); J_3(S_{ij}); \alpha^m) = G^S(I_2(S'_{ij}); I_3(S'_{ij}); \alpha^m) = K \tag{4}$$

Traditional procedures for deriving the constitutive equations for anisotropic elastoplastic materials are based on the description of appropriate yield and potential functions in terms of the characteristic material properties<sup>21,22</sup>. Satisfaction of the invariance condition in these cases is difficult and not always possible. A procedure to guarantee this condition proposed in this work is to define the properties of the real anisotropic solid in terms of those for a fictitious isotropic solid<sup>1,2</sup>. This is achieved by relating the stresses in the real and fictitious spaces using the following linear transformation (see Figure 1):

$$\bar{S}_{ij} = A^S_{ijkl} S_{kl} \tag{5}$$

where  $S_{ij}$  and  $\bar{S}_{ij}$  are the stress tensor in the real anisotropic solid and the fictitious isotropic solid, respectively, and  $A^S_{ijkl}$  is a fourth order material tensor, termed stress space transformation tensor defined as a ratio of the material strength, i.e.:

$$A^S_{ijkl} = \frac{f^{\bar{S}}_{ij}}{f^S_{ij}} \tag{6}$$

where  $f^S_{ij}$  and  $f^{\bar{S}}_{ij}$  are the yield strength tensors corresponding to the real and fictitious solids, respectively.

To ensure non-proportionality between the strength and the elastic modulus the following relation between the real elastic strains  $E^e_{ij}$  and the fictitious ones  $\bar{E}^e_{ij}$  is defined:

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This assumption implies non-uniqueness of elastic strains when the change of space is produced. In (7)  $A^E_{ijkl}$  is a fourth order material tensor, termed strain space transformation tensor. This can be derived from (5) as follows:

$$\bar{S}_{ik} S^{-1}_{jl} = A^S_{ijkl} = (C^{\bar{S}}_{ikrs} \bar{E}^e_{rs}) (C^S_{jlmn} E^e_{mn})^{-1}$$

$$= C^{\bar{S}}_{ikrs} \underbrace{\bar{E}^e_{rs} E^e_{mn}^{-1}}_{A^E_{rsmn}} C^{S\ x}_{mnjl}^{-1}$$

$$C^{\bar{S}}_{ikrs} A^E_{rsmn} C^S_{mnjl}^{-1} \tag{8a,b}$$

and hence:

$$A^E_{rsmn} = C^{\bar{S}}_{rsik}^{-1} A^S_{ijkl} C^S_{jlmn}$$

Equation (8b) allows us to derive the relationship between the constitutive tensors in the real

$C^S_{ijkl}$  and fictitious  $C^{\bar{S}}_{ijkl}$  spaces. This is:

$$I_{rstu} = A^E_{rsmn} A^E_{mntu}^{-1} = C^{\bar{S}}_{rsik}^{-1} \underbrace{A^S_{ijkl} C^S_{jlmn} A^E_{mntu}^{-1}}_{C^{\bar{S}}_{iktu}} \tag{9}$$

or the inverse relation:

$$C^S_{jlmn} = A^S_{klij}^{-1} C^{\bar{S}}_{iktu} A^E_{tu mn}$$

Note that both  $C^S_{ijkl}$  and  $f^S_{ij}$  are expressed in a global reference system. This means that prior to the derivation of  $A^S_{ijkl}$  and  $A^E_{ijkl}$  the following transformations to the global reference system are required:

$$C^S_{ijkl} = R_{irjs} (C^{rs pq})_{local} R_{kplq}$$

$$f^S_{ij} = R_{ijkl} (f^{kl})_{local}$$

where  $(\cdot)_{local}$  denotes description in a local coordinate system.

The rotation tensor\*  $R_{ijkl}$  takes into account the angles between the local principal directions of the anisotropic material and those of the global coordinate system. Superindices  $S$  and  $\bar{S}$  denote hereafter variables in the real (anisotropic) and fictitious (isotropic) spaces, respectively.

The mapping expressed by means of (5), induces a change in the yield function shape as can be seen in Figure 1. Figure 2, shows this effect for different strength ratios ( $S_{ii}/S_{ii}$ ) on four classical yield functions<sup>14,16</sup>: (A) Tresca, (B) Von Mises, (C) Mohr–Coulomb, (D) Drucker–Prager, and also on that proposed by (E) Lubliner and Oller<sup>13,16</sup>. This space mapping allows the representation of the one directional fibre yield function when the ratio  $S_{ii}/S_{ii}$  tends to infinity. Figures 2 and 3, show for associated plasticity, the loss of the strength in a given direction while in the normal one there is a plastic flow growth in the same proportion.

Assuming that all the information concerning material anisotropy is contained in the tensor  $A^S_{ijkl}$ , the yield and plastic potential functions for the anisotropic solid are defined as:

$$F^S(S_{ij}; \alpha^m_S) = F^{\bar{S}}(S_{ij}; A^S_{ijkl}; \alpha^m_S) = F^{\bar{S}}(\bar{S}_{ij}; \alpha^m_{\bar{S}}) = 0$$

$$G^S(S_{ij}; \alpha^m_S) = G^{\bar{S}}(S_{ij}; A^S_{ijkl}; \alpha^m_S) = G^{\bar{S}}(\bar{S}_{ij}; \alpha^m_{\bar{S}}) = K \tag{10}$$

It is therefore concluded that the yield and plastic potential functions for any anisotropic solid can be simply defined in terms of an irreducible basis of the invariants of tensor  $S_{ij}$ . Usually, a finite number of stress invariants are involved in the definition of the yield and potential functions (i.e.:  $I_1, J_2$  and sometimes  $J_3$ ). These invariants are elements of the system of invariants which are only considered. Therefore the formulation presented is a simplified theory but very useful to solve practical problems.

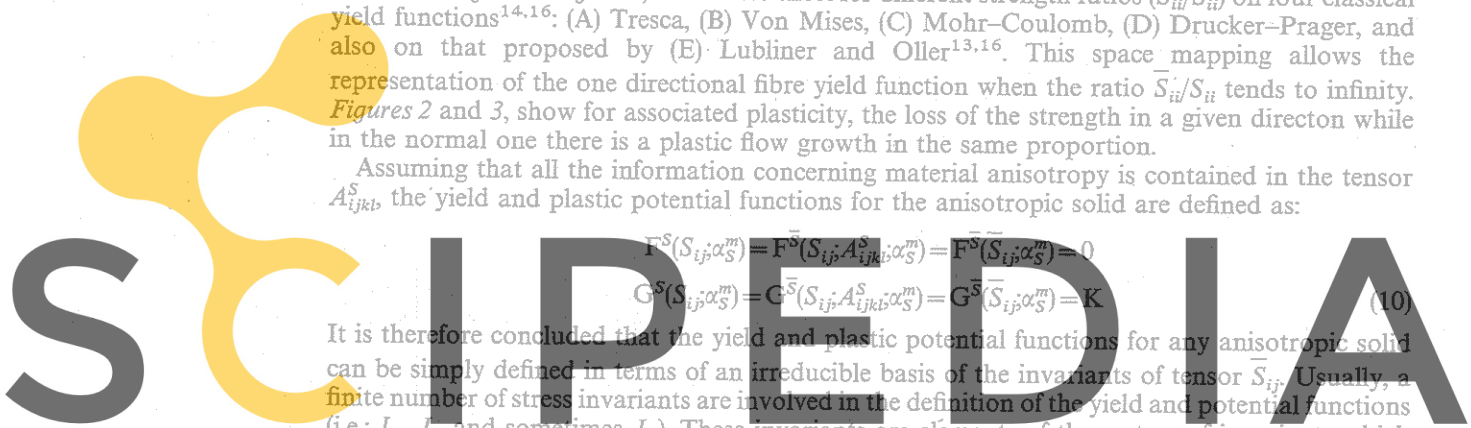
In the following sections the main relationships characterizing an anisotropic elastoplastic solid formulated as an ideal isotropic solid using the spaces transformation expressed by (5) and (7) are derived.

**Secant constitutive equation**

The constitutive equation for an anisotropic material is obtained by writing the dissipation occurring in an isothermic elastoplastic process in the real anisotropic space. From the first Classius–Planck condition<sup>10–12,14,17,18</sup> the following expression for the mechanical dissipation power is obtained:

$$\dot{E}^S_{mec} = \left( \frac{S_{ij}}{m^0} - \frac{\partial \Psi^S}{\partial E^e_{ij}} \right) \dot{E}^e_{ij} + \frac{S_{ij} \dot{E}^p_{ij}}{m^0} - \sum_m \frac{\partial \Psi^S}{\partial \alpha^m_S} \dot{\alpha}^m_S \geq 0 \tag{11}$$

\*The rotation tensor definition is:  $R_{ijkl} = r_{ik} r_{jl}$ , where  $r_{ij} = \cos((\bar{e}_i)_{global}, (\bar{e}_j)_{local})$ , and  $(\bar{e}_i)_{reference}$  is the unit vector corresponding to the  $k$  component of a certain reference coordinate system.



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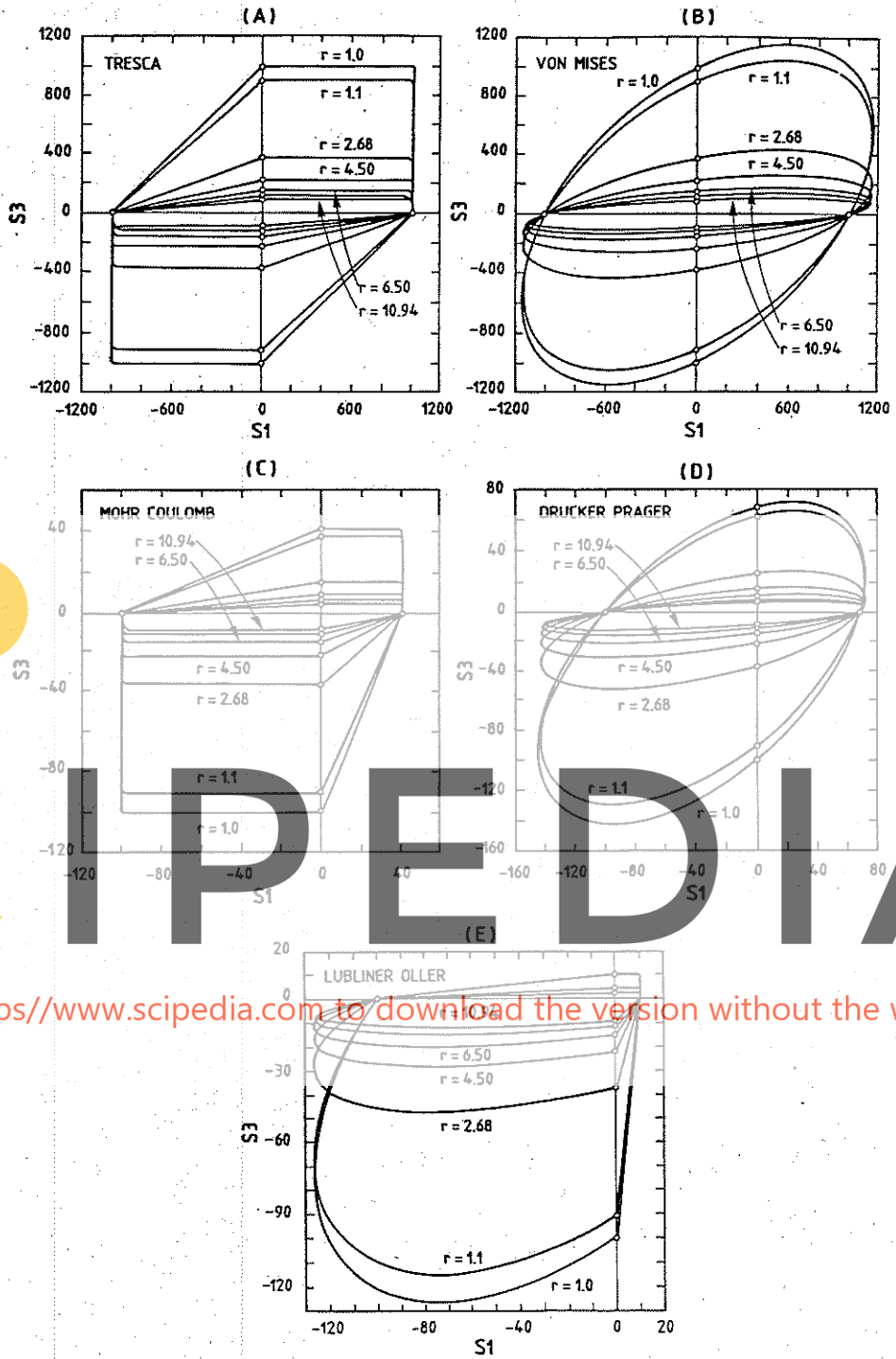


Figure 2 Shape effect for different strength ratios ( $\bar{S}_{ii}/S_{ii}$ ) on several yield functions: (A) Tresca, (B) Von Mises, (C) Mohr-Coulomb, (D) Drucker-Prager, and (E) Lubliner-Oller



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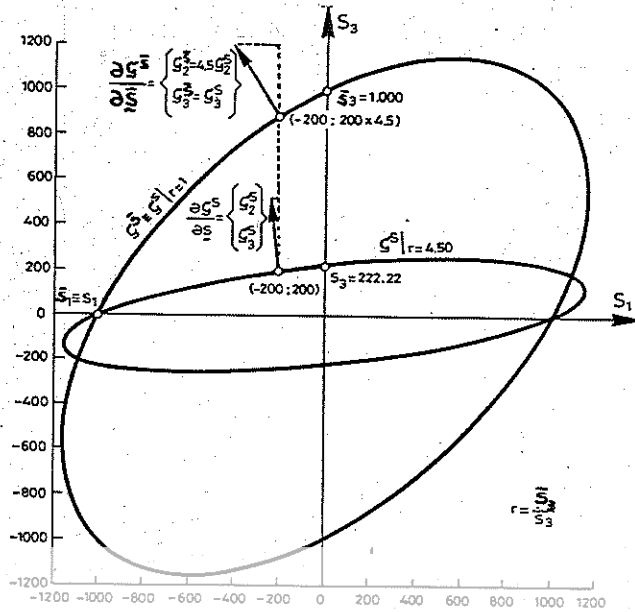


Figure 3 Differences in the plastic flow on the real ( $r=1$ ) and fictitious spaces ( $r=4.5$ )

Applying Coleman's method<sup>11</sup> to guarantee the condition of positive dissipation in (11), the secant constitutive equation is obtained as:

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$$S_{ij} = m^0 \frac{\partial \Psi^S(E_{ij}^e; \alpha_s^m)}{\partial E_{ij}^e} \quad (12)$$

where  $\Psi^S$  is the free energy of material formulated in the material configuration under real stress state and  $m^0$  the density in the material configuration. The free energy is assumed to be of the form:

$$\Psi^S(E_{ij}^e; \alpha_s^m) = \Psi^{S^e}(E_{ij}^e) + \Psi^{S^p}(\alpha_s^m) = \left( \frac{1}{2m_0} E_{ij}^e C^S_{ijkl} E_{kl}^e \right) + \Psi^{S^p}(\alpha_s^m) \quad (13)$$

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where  $\Psi^{S^e}$  and  $\Psi^{S^p}$  denote the elastic and plastic free energy contributions, and the free variable  $E_{ij}^e$  is the fictitious elastic strain in the material configuration as proposed by Green-Naghdi<sup>6,11,12</sup> and defined as:

$$E_{ij}^e = E_{ij} - E_{ij}^p = \frac{1}{2}(C_{ij} - I_{ij}) - \int_0^r \dot{E}_{ij}^p dt \quad (14)$$

where  $E_{ij}^p$  is the Lagrangian plastic strain,  $C_{ij}$  is the right Cauchy-Green tensor and  $\dot{E}_{ij}^p$  the plastic strain rate defined in the material configuration as explained in the next section.

Substitution of (9) into (13) allows us to rewrite the free energy in terms of the constitutive tensor  $C^S_{ijkl}$  for the ideal isotropic material as:

$$\Psi^S(E_{ij}^e; \alpha_s^m) = \left( \frac{1}{2m_0} E_{ij}^e [A^S_{ijpq}]^{-1} C^S_{pqur} A^E_{turs} E_{rs}^e \right) + \Psi^{S^p}(\alpha_s^m) \quad (15)$$

Substituting (15) into (12) leads to the secant constitutive equation in the real anisotropic space defined in terms of the stress field in the ideal isotropic fictitious space as:

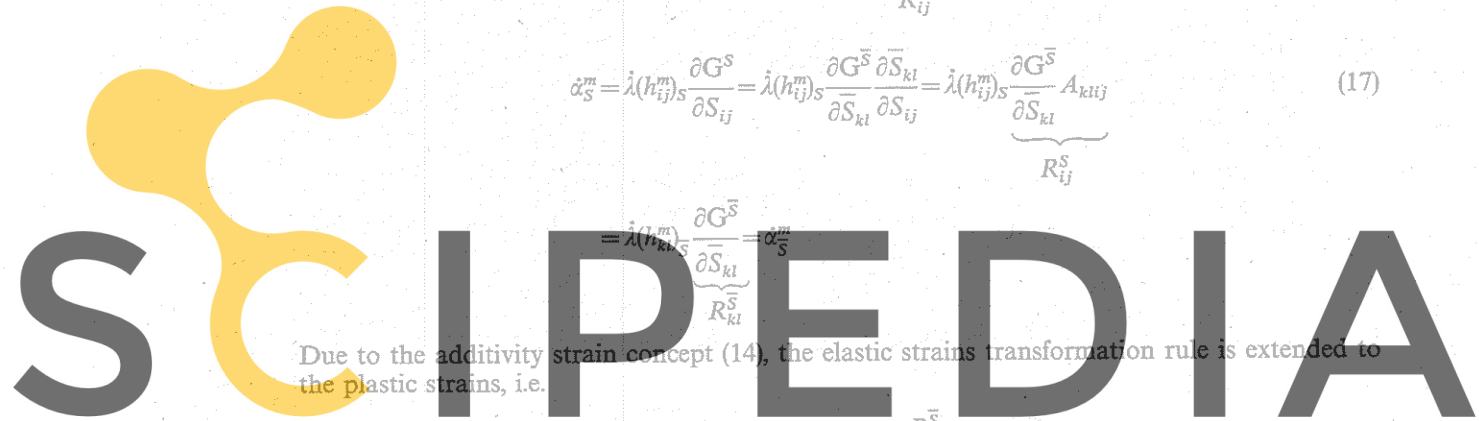
$$S_{ij} = m^0 \frac{\partial \Psi^S(E_{ij}^e, \alpha_S^m)}{\partial E_{ij}^e} = [A_{ijpq}^S]^{-1} C_{pqtu}^S [A_{turs}^E] E_{rs}^e = A_{ijpq}^S]^{-1} C_{pqtu}^S \bar{E}_{tu}^e = A_{ijpq}^S]^{-1} \bar{S}_{pq} \quad (16)$$

where  $\bar{S}_{ij} = C_{ijkl}^S \bar{E}_{kl}^e$  are the stresses in the fictitious isotropic space. Equation (16) confirms the assumptions made in (5) and (7).

*Flow rule. Evolution of the internal variables*

From the transformation rules expressed by (5) and (7) and the definition of the plastic potential function in the fictitious isotropic space ((10)), the flow rule and the evolution of the internal plastic variables  $\alpha_S^m$  are obtained in the form:

$$\begin{aligned} \dot{E}_{ij}^p &= \lambda \frac{\partial G^S}{\partial S_{ij}} = \lambda \frac{\partial G^S}{\partial \bar{S}_{kl}} \frac{\partial \bar{S}_{kl}}{\partial S_{ij}} = \lambda \frac{\partial G^S}{\partial \bar{S}_{kl}} \underbrace{A_{kl ij}^S}_{R_{ij}^S} = (\dot{E}_{kl}^p) \bar{S}_{kl} A_{kl ij}^S \\ \dot{\alpha}_S^m &= \lambda (h_{ij}^m)_S \frac{\partial G^S}{\partial S_{ij}} = \lambda (h_{ij}^m)_S \frac{\partial G^S}{\partial \bar{S}_{kl}} \frac{\partial \bar{S}_{kl}}{\partial S_{ij}} = \lambda (h_{ij}^m)_S \frac{\partial G^S}{\partial \bar{S}_{kl}} \underbrace{A_{kl ij}^S}_{R_{ij}^S} \end{aligned} \quad (17)$$



Due to the additivity strain concept (14), the elastic strains transformation rule is extended to the plastic strains, i.e.

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where  $(h_{ij}^m)_S$  and  $(h_{ij}^m)_{\bar{S}}$  are tensorial functions to be determined<sup>13,16</sup> for each of the  $m$  internal variables involved,  $R_{kl}^S$  is the plastic flow in the fictitious stress space, as shown in Figure 3,  $R_{ij}^S$  is the plastic flow in the real stress space,  $R_{rs}^S$  is the plastic flow in the fictitious stress space,  $R_{kl}^{\bar{S}}$  is the plastic flow in the fictitious isotropic space and  $\bar{E}_{rs}^p$  is the fictitious plastic strain.

*Free energy in the fictitious isotropic space. Uniqueness of the dissipation*

Note that the dissipated mechanical power  $\Xi_{mec}^S$  can be written in the fictitious isotropic space  $\Xi_{mec}^{\bar{S}}$  by substituting the flow rule, the evolution law for the internal variables ((17)) and the

transformation space rules ((5) and (7)) into (10), i.e.\*:

$$\begin{aligned}\Xi_{\text{mec.}}^S &= \left( \frac{S_{ij}}{m^0} \frac{\partial \Psi^S}{\partial E_{ij}^e} \right) \dot{E}_{ij}^e + \frac{S_{ij} \dot{E}_{ij}^p}{m^0} - \sum_m \frac{\partial \Psi^S}{\partial \alpha_S^m} \dot{\alpha}_S^m \\ &= \frac{\bar{S}_{ij} (\dot{E}_{ij}^p)^{\bar{S}}}{m^0} - \sum_m \frac{\partial \Psi^{\bar{S}}}{\partial \alpha_S^m} \dot{\alpha}_S^m \equiv \Xi_{\text{mec.}}^{\bar{S}} \geq 0\end{aligned}\quad (19)$$

It is deduced from (19) that the dissipation is an invariant of the thermodynamic process and, therefore, its value is independent of the space where it is computed. also, the free energy can be obtained in the fictitious isotropic space in the form:

$$\Psi^{\bar{S}}(\bar{E}_{ij}^e; \alpha_S^m) = \left( \frac{1}{2m_0} \bar{E}_{ij}^e C_{ijkl}^{\bar{S}} \bar{E}_{kl}^e \right) + \Psi^{\bar{S}^p}(\alpha_S^m) \quad (20)$$

Equation (20) is equivalent to (13) defined in the real anisotropic space, and therefore the constitutive equation given by (16) still holds. This is:

$$\begin{aligned}\bar{S}_{ij} &= m^0 \frac{\partial \Psi^{\bar{S}}(\bar{E}_{ij}^e; \alpha_S^m)}{\partial \bar{E}_{ij}^e} = C_{ijkl}^{\bar{S}} \bar{E}_{kl}^e \\ &= (A_{ijrs}^{\bar{S}} C_{rstu}^{\bar{S}} A_{tukl}^E) (A_{klpq}^E E_{pq}^e) \\ &= A_{ijrs}^{\bar{S}} C_{rstu}^{\bar{S}} I_{tupq} E_{pq}^e \\ &= A_{ijrs}^{\bar{S}} C_{rstu}^{\bar{S}} E_{tu}^e = A_{ijrs}^{\bar{S}} S_{rs}\end{aligned}\quad (21)$$

where the stress transformation rule is recovered.

Tangent constitutive equation

The rate form of the constitutive equation is obtained by performing the temporal derivative

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\*REMARK: The plastic potential transformation is derived in the following manner:

$$\begin{aligned}S_{ij} \dot{E}_{ij}^p &= (A_{ijrs}^{\bar{S}})^{-1} \bar{S}_{rs} (A_{ijkl}^E)^{-1} \dot{E}_{kl}^p \\ &= \lambda A_{ijrs}^{\bar{S}} {}^{-1} \bar{S}_{rs} A_{ijkl}^E {}^{-1} A_{kltu}^E \frac{\partial G^{\bar{S}}}{\partial \bar{S}_{np}} A_{nptu}^{\bar{S}} \\ &= \lambda A_{ijrs}^{\bar{S}} {}^{-1} \bar{S}_{rs} I_{ijtu} \frac{\partial G^{\bar{S}}}{\partial \bar{S}_{np}} A_{nptu}^{\bar{S}} \\ &= \lambda A_{ijrs}^{\bar{S}} {}^{-1} \bar{S}_{rs} \frac{\partial G^{\bar{S}}}{\partial \bar{S}_{np}} A_{npij}^{\bar{S}} = \lambda \bar{S}_{rs} \frac{\partial G^{\bar{S}}}{\partial \bar{S}_{np}} I_{nprs} \\ &= \lambda \bar{S}_{np} \frac{\partial G^{\bar{S}}}{\partial \bar{S}_{np}} = \bar{S}_{np} (\dot{E}_{ij}^p)^{\bar{S}}\end{aligned}$$



of the secant expression (16) as:

$$\begin{aligned} \dot{S}_{ij} &= \frac{\partial S_{ij}}{\partial E_{kl}^e} \dot{E}_{kl}^e = \frac{\partial S_{ij}}{\partial S_{rs}} \underbrace{\left( \frac{\partial S_{rs}}{\partial E_{mn}^e} \frac{\partial E_{mn}^e}{\partial E_{kl}^e} \right)}_{\dot{\bar{S}}_{rs}} \dot{E}_{kl}^e \\ &= A_{ijrs}^{-1} \underbrace{C_{rsmn}^S}_{A_{ijrs}^{S^{-1}}} \underbrace{A_{mnlk}^E}_{C_{rsmn}^S} \dot{E}_{kl}^e = A_{ijrs}^{-1} C_{rsmn}^S A_{mnlk}^E (\dot{E}_{kl}^e - \dot{E}_{kl}^p) \\ &= A_{ijrs}^{-1} \underbrace{C_{rsmn}^S}_{\dot{\bar{S}}_{rs}} (\dot{E}_{mn}^e - \dot{E}_{mn}^p) \end{aligned} \quad (22)$$

Equation (22) can be interpreted as a linear transformation of the following rate constitutive equation defined in the fictitious isotropic space:

$$\dot{\bar{S}}_{ij} = \frac{\partial \bar{S}_{ij}}{\partial \bar{E}_{kl}^e} \dot{\bar{E}}_{kl}^e = C_{ijkl}^{\bar{S}} \dot{\bar{E}}_{kl}^e = C_{ijkl}^{\bar{S}} (\dot{\bar{E}}_{kl}^e - \dot{\bar{E}}_{kl}^p) \quad (23)$$

The plastic consistency condition leads to the standard rate form of the constitutive equation in the fictitious isotropic space as:

$$\dot{\bar{S}}_{ij} = \left\{ C_{ijkl}^{\bar{S}} \frac{\left( C_{ijrs}^{\bar{S}} R_{rs}^{S^*} \right) \left( \frac{\partial F^{\bar{S}}}{\partial S_{rs}} C_{rskl}^{\bar{S}} \right)}{-\sum_m \frac{\partial F^{\bar{S}}}{\partial \alpha_S^m} (h_{tu}^m)_S R_{tu}^{\bar{S}} + \frac{\partial F^{\bar{S}}}{\partial S_{pq}} C_{pqln}^{\bar{S}} R_{ln}^{S^*}} \right\} \dot{\bar{E}}_{kl}^e \quad (24a)$$

or in compact form:

$$\dot{\bar{S}}_{ij} = C_{ijkl}^{\bar{S}^*} \dot{\bar{E}}_{kl}^e \quad (24b)$$

Combining now (22) and (23) leads to the final expression of the rate constitutive equation in the real anisotropic solid as:

$$\dot{S}_{ij} = A_{ijkl}^{-1} \dot{\bar{S}}_{kl} = A_{ijkl}^{S^{-1}} C_{klrs}^{\bar{S}^*} \dot{E}_{rs} = A_{ijkl}^{S^{-1}} C_{klrs}^{\bar{S}^*} A_{rstu}^E \dot{E}_{tu}^e$$

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Therefore, the constitutive model requires only the following definitions in each of the two spaces used:

- *real anisotropic space:*
  - { initial constitutive tensor  $(C_{ijkl}^S)_{local}$
  - { yield strength  $(f_{ij}^S)_{local}$
- *fictitious isotropic space:*
  - { initial fictitious constitutive tensor  $C_{ijkl}^{\bar{S}}$
  - { yield function  $F^{\bar{S}}(\bar{S}_{ij}, \alpha_S^m) = 0$
  - { potential function  $G^{\bar{S}}(\bar{S}_{ij}, \alpha_S^m) = K$
  - { yield strength  $(f_{ij}^{\bar{S}})$

Above definitions allow us to derive all the basic constitutive relationships, including the evolution of the inner variables, necessary to formulate in a precise manner the constitutive behaviour of an anisotropic elastoplastic solid by means of an associated fictitious isotropic solid.

## INTEGRATION OF THE CONSTITUTIVE EQUATION

The constitutive equation is integrated by means of a tangent cutting plane algorithm as presented by Ortiz and Simo<sup>18</sup>. This kind of Euler-backward algorithm is used for integration of rate constitutive equations in orthotropic solids<sup>5</sup>. Nevertheless, in our case, note that the integration of the constitutive equation is done on the isotropic fictitious space. The integration methodology is based in the 'mapping return' concept, i.e. the return path towards the converged yield function follows the direction of the normal flow computed at each iterative solution (see *Figure 4*). Further details of this technique can be found in References 4, 5, 19.

### NUMERICAL IMPLEMENTATION OF THE ANISOTROPIC ELASTOPLASTIC MODEL PROPOSED

The basic steps for implementing the anisotropic model proposed into standard elastoplastic finite element programs<sup>24</sup> are given next.

- LOOP OVER SOLUTION INCREMENT:  $n=1$ ,  $NINCR$
- ITERATION LOOP:  $i=1$ ,  $NITER$
- IF ( $n>GT.1$  OR  $i>GT.1$ ) GOTO 4

1. Define stiffness and rotation tensors:

$$\bar{f}^S; (f^S)_{local}; \bar{C}^S; (C^S)_{local}; R$$

2. Obtain the spaces transformation tensors:

$$f^S = R: (f^S)_{local}$$

$$A^S = \bar{f}^S \otimes f^S$$

$$C^S = R: (C^S)_{local}; R$$

$$A^E = C^{S^{-1}}: A^S; C^S$$

3. Initialize the isotropic constitutive tensor:

$$I(C^{\bar{S}^{ep}0}) = I(C^{\bar{S}^0})$$

4. Compute tangent stiffness:

$${}^n(C^{S^{ep}i-1}) = A^{S^{-1}}: {}^n(C^{\bar{S}^{ep}i-1}): A^E$$

$${}^n(K^{(e)i-1}) = \int_V B: {}^n(C^{S^{ep}i-1}): B dV$$

$${}^n(K)^{-1} = A_{e=1}^{ne} : {}^n(K^{(e)})^{-1}$$

5. Compute displacements and strains:

$${}^n(\delta U)^i = {}^n(K^{-1})^{i-1} : {}^n(F_{resid})^{i-1}$$

$${}^n(\Delta U)^i = {}^n(\Delta U)^{i-1} + {}^n(\delta U)^i$$

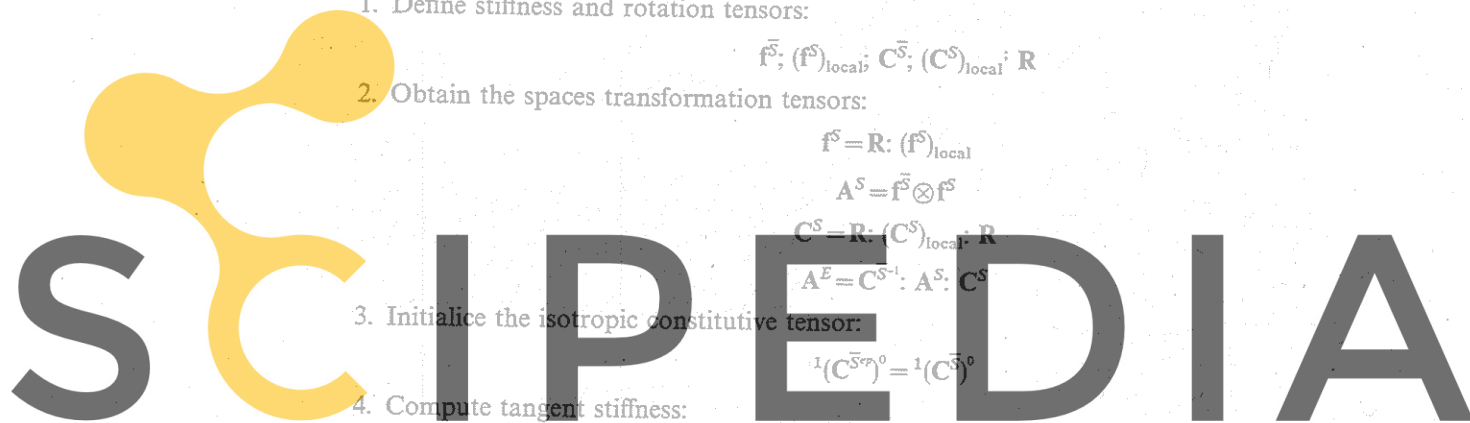
$${}^n(\Delta E)^i = f^i({}^n(\Delta U)^i)$$

$${}^n(E)^i = {}^{n-1}(E) + {}^n(\Delta E)^i$$

6. Evaluate predicted stresses:

$${}^n(\Delta S^*)^i = I(C^{\bar{S}^0}): {}^n(\Delta E)^i$$

$${}^n(S^*)^i = {}^{n-1}(S) + {}^n(\Delta S^*)^i$$



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7. Transform predicted stresses to the fictitious isotropic space:

$${}^n(\bar{S})^i = A^S : {}^n(S^*)^i$$

8. Integrate isotropic constitutive law: ITERATION LOOP:  
 $j=1$ , JITER

$$\text{for: } j=1 \Rightarrow {}^n(\bar{S})^{i0} = {}^n(\bar{S}^*)^i; \quad {}^n(\Delta \bar{E}^p)^{i0} = 0$$

• 8.1

$${}^n(\bar{S})^{ij} = {}^n(\bar{S})^{ij-1} - {}^1(C^S)^0 : {}^n(\Delta \bar{E}^p)^{ij-1}$$

$$\text{If: } F^{\bar{S}}(\bar{S}; \alpha_{\bar{S}}^m)_{n\bar{S}^{ij-1}} \leq 0 \quad \begin{cases} \text{and } j=1 \Rightarrow \text{GOTO 10} \\ \text{and } j > 1 \Rightarrow \text{GOTO 9} \end{cases}$$

else:

$${}^n(\Delta \bar{E}^p)^{ij} = (\Delta \lambda)^j \cdot A^E : \left( \frac{\partial G^{\bar{S}}}{\partial S} \right)_{n\bar{S}^{ij}}^j : A^S = (\Delta \lambda)^j \cdot {}^n(R)_{n\bar{S}^{ij}}^j$$

$${}^n(\bar{E}^p)^{ij} = {}^n(\bar{E}^p)^{ij-1} + {}^n(\Delta \bar{E}^p)^{ij}$$

$${}^n(\Delta \alpha_{\bar{S}}^m)^{ij} = (\Delta \lambda)^j \cdot {}^n(h_{\bar{S}}^m)^{ij} : \left( \frac{\partial G^{\bar{S}}}{\partial S} \right)_{n\bar{S}^{ij}}^j$$

$${}^n(\alpha_{\bar{S}}^m)^{ij} = {}^n(\alpha_{\bar{S}}^m)^{ij-1} + {}^n(\Delta \alpha_{\bar{S}}^m)^{ij}$$

$k=k+1$  Go back to 8.1 •

9. Compute tangent constitutive tensor in the isotropic space:

$${}^n(C^{\bar{S}^*})^i = \left\{ C^{\bar{S}} - \frac{(C^{\bar{S}} : R^*) \otimes \left( \frac{\partial F^{\bar{S}}}{\partial S} C^{\bar{S}} \right)}{-\sum_m \frac{\partial F^{\bar{S}}}{\partial \alpha_{\bar{S}}^m} (h_{\bar{S}}^m) : R^* + \frac{\partial F^{\bar{S}}}{\partial S} C^{\bar{S}} : R^*} \right\}_{n\bar{S}^i}$$

10. Back transformation of stresses to the real anisotropic space:

$${}^n(S)^i = A^{S^{-1}} : {}^n(\bar{S})^i$$

11. Compute residual forces:

$${}^n(F_{\text{resid}}^{(e)})^i = \int_V B : {}^n(S)^i dV - f_{\text{ext}}$$

$${}^n(F_{\text{resid}}) = A_{e=1}^{ne} : {}^n(F_{\text{resid}}^{(e)})^i$$

If  $\|F_{\text{resid}}\| > 0? \Rightarrow i=i+1$  Go back to 4

else:

Converged solution for the  $n$ th increment.

$$n = n + 1$$

Compute new incremental solution

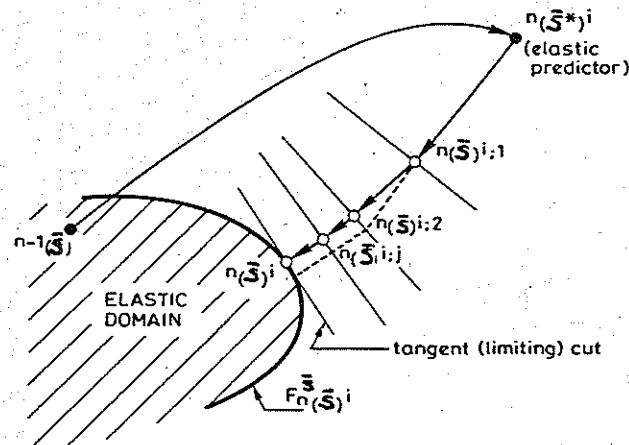


Figure 4 Plastic flow definition in the cutting plane algorithm<sup>18</sup>

## EXAMPLES

### Orientation effect of fibres in a composite

Let us consider the case where continuous glass fibres are placed uni-directionally within an epoxy matrix material and form an angle  $\phi$  with the applied tension load as shown in *Figure 5*. The tension failure limit strength ( $S_y$ ) of the composite depends on the fibre orientation which angle will be varied between  $0^\circ$  and  $90^\circ$ . In order to establish strength and failure characteristics of the material, it is necessary to consider a number of fundamental properties related to the composite behaviour such as: Young's modulus along principal directions: in-plane directions,  $E_{\text{Long}} = 591\,998.8 \text{ kg/cm}^2$ , and  $E_{\text{Trans}} = 140\,617.3 \text{ kg/cm}^2$ . Poisson ratio:  $\nu_{LT} = 0.293$ . Elastic strength limits, in-plane directions: for the longitudinal fibre behaviour  $f_{\text{Long}}^{\text{Comp.}} = 19\,686.4 \text{ kg/cm}^2$ ,  $f_{\text{Long}}^{\text{Tens.}} = 9561.9 \text{ kg/cm}^2$ ,  $f_{LT}^{\text{Shear.}} = 421.8 \text{ kg/cm}^2$ ; and for the transverse fibre behaviour  $f_{\text{Trans}}^{\text{Comp.}} = 1406.2 \text{ kg/cm}^2$ ,  $f_{\text{Trans}}^{\text{Tens.}} = 281.2 \text{ kg/cm}^2$ .

Additional data required defining the fictitious isotropic model are: equivalent Young's modulus:  $E^{\bar{S}} = 591\,998.8 \text{ kg/cm}^2$ , equivalent elastic compression strength limit;  $f^{\text{au}} = 19\,686.4 \text{ kg/cm}^2$ , Mohr–Coulomb yield function, fibre plastic flow (it only has components on the fibre direction) and perfect plasticity. *Figure 5* shows the tension strength limit ( $S_y/f_T$  where  $f_T$  is the tension failure limit strength of the fibres) of the composite as the fibre orientations angle  $\phi$  varies. Note that the maximum strength takes place for  $\phi = 0^\circ$  and the minimum for  $\phi = 45^\circ$ . The material strength is greater for  $\phi = 90^\circ$  than for  $\phi = 45^\circ$  due to the transversal deformation restricted by the compression fibre strength of the fibres.

The particular form of the curve in *Figure 5*, depends on the yield and potential plastic functions adopted in the fictitious isotropic solid. Here the Mohr–Coulomb yield function was arbitrarily chosen.

### A comparison with the Mises–Hill orthotropic model

The anisotropic model proposed has been applied to the analysis of a rectangular specimen under axial loading acting along three different directions (*Figure 6*). Plane stress conditions have been assumed. The geometry has been discretized using a simple mesh of sixteen standard 4 nodes quadrilateral elements as shown in *Figure 6*. Numerical results obtained have been compared with those provided by the well known orthotropic Mises–Hill model<sup>7</sup>.

The material properties are the following: Young's modulus along principal directions: in-plane directions,  $E_1 = 2.5 E^6 \text{ kg/cm}^2$ , and  $E_2 = 2.0 E^6 \text{ kg/cm}^2$ . Transverse direction  $E_3 = 2.0 E^6 \text{ kg/cm}^2$ .

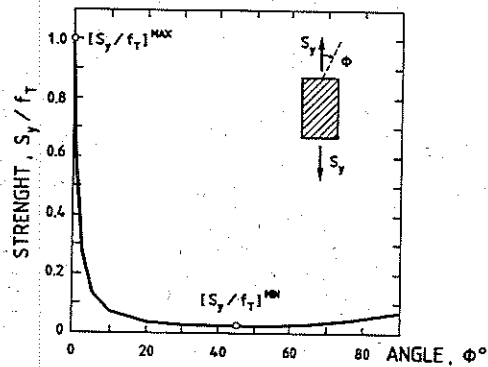


Figure 5 Strength ratio vs. fibre slope angle, for an epoxy matrix with glass fibre

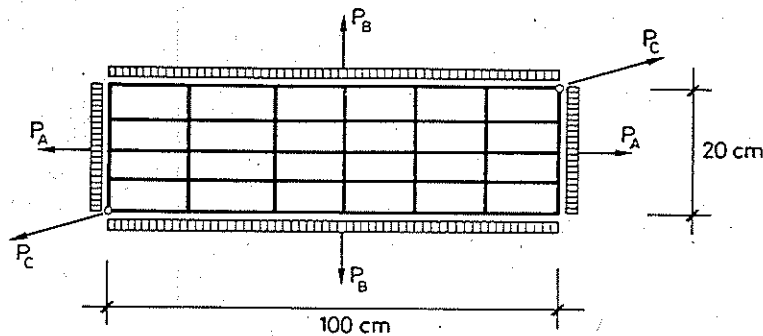


Figure 6 Test specimen. Geometry, boundary condition, loading and finite element

Shear modulus:  $G = 0.45 E^6 \text{ kg/cm}^2$ . Poisson ratio:  $\nu = 0.3$ . Elastic strength limits: in-plane directions;  $f_1 = 8000.0 \text{ kg/cm}^2$  and  $f_2 = 6000.0 \text{ kg/cm}^2$ , transverse direction  $f_3 = 6000.0 \text{ kg/cm}^2$ .

Additional data defining the fictitious isotropic model are: equivalent Young's modulus:  $E^S = 2.5 E^6 \text{ kg/cm}^2$ , equivalent elastic strength limit:  $f^{au} = 8000.0 \text{ kg/cm}^2$ , and Von-Mises yield function.

Figure 6 shows the three loading types applied: (a) axial loading parallel to the maximum strength direction; (b) transverse loading orthogonal to the maximum strength direction, and (c) diagonal loading.

Figures 7 and 8 show the comparison between the strain-stress curves obtained at the centre of the sample, assuming perfect plasticity and hardening plasticity with a hardening modulus  $H = 250\,000.0 \text{ kg/cm}^2$ , respectively.

Note that results obtained with both models coincide for loading cases (a) and (b). However, considerable differences arise for the diagonal loading case (c). This is due to the excessive influence of shear in Mises-Hill theory leading to over-stiff results. The values obtained with the proposed model are within reasonable limits and bounded by those of loading cases (a) and (b) as expected.

#### Analysis of a fibre-reinforced composite

The third example is the analysis of a plane rectangular specimen of a fibre-reinforced composite material. Figure 9 shows the specimen geometry, the boundary conditions and the finite element

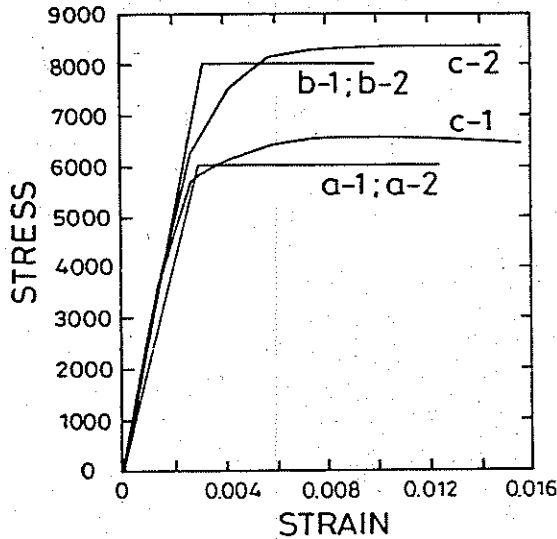


Figure 7 Strain-stress curves at centre of the sample for perfect plasticity using the anisotropic model proposed and the Mises-Hill model. Loading (a): (a-1) present model, (a-2) Mises-Hill model; Loading (b): (b-1) present model, (b-2) Mises-Hill model; Loading (c): (c-1) present model, (c-2) Mises-Hill model

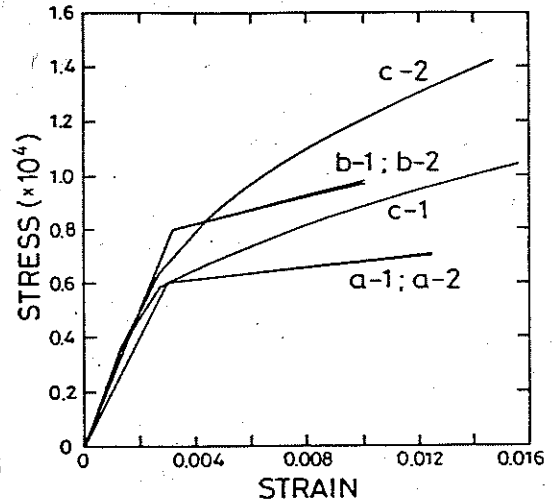


Figure 8 Strain-stress curves at centre of the sample for hardening plasticity case using the anisotropic model proposed and the Mises-Hill model. Loading (a): (a-1) present model, (a-2) Mises-Hill model; Loading (b): (b-1) present model, (b-2) Mises-Hill model; Loading (c): (c-1) present model, (c-2) Mises-Hill model

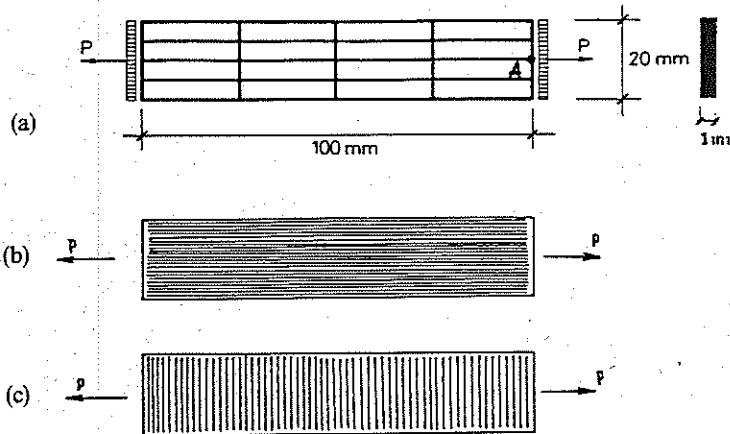


Figure 9 Fibre-reinforced specimen. Geometry, boundary condition, loading and finite element mesh

mesh of sixteen 4 node quadrilateral elements used. A prescribed longitudinal displacement has been imposed to both ends of the specimen as shown in Figure 9.

The specimen is composed of an isotropic matrix and a 20% of long fibres. Initially, the fibres are considered aligned along the longitudinal direction and then along the transverse direction. The combined effect of fibre and matrix material has been modelled using a multiphase model based on mixing theory developed by the authors<sup>25</sup>.

*Material properties:* Matrix isotropic material:  $E = 7.24 \text{ tn/mm}^2$ ,  $\nu = 0.33$ , limit elastic strength

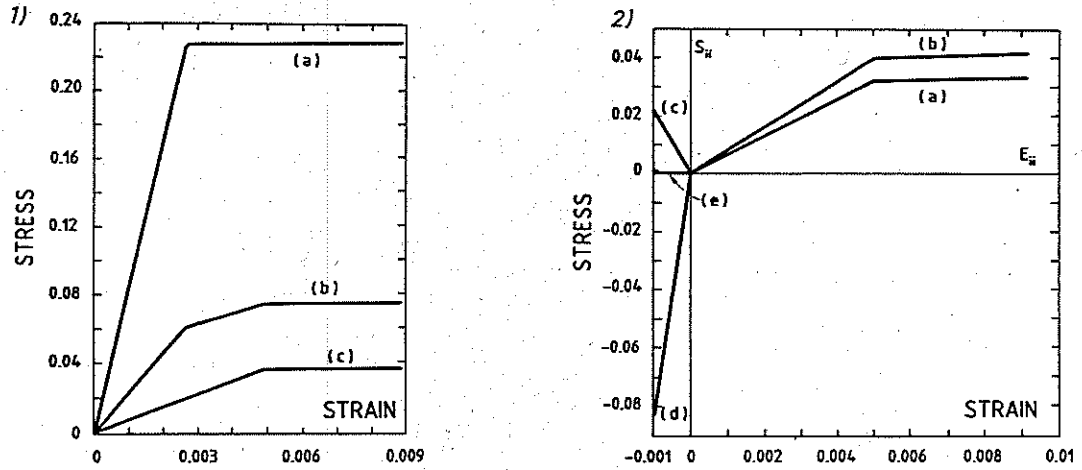


Figure 10 Fibre-reinforced composites: stress-strain curves in the centre of the specimen. Case (1): longitudinal fibres. Behaviour on the longitudinal direction of the: (a) fibres; (b) composite; (c) matrix. Case (2): transverse fibres. Behaviour on the longitudinal direction of the: (a) composite; (b) matrix. Behaviour on the transverse direction of the: (c) matrix; (d) fibres; (e) composite

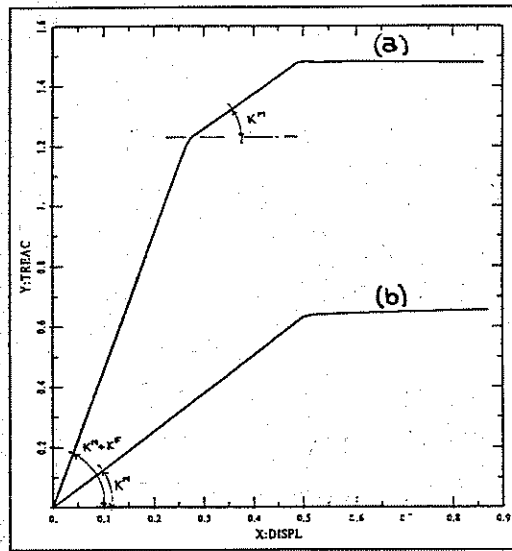


Figure 11 Load-displacement curves in point 'A' of the specimen. (a): longitudinal fibres; (b): transverse fibres

$f=0.036 \text{ tn/mm}^2$  and 80% of fraction volume participation. Isotropic associated Von Mises plasticity has been assumed in this case.

*Fibres.* The anisotropic model proposed has been used with  $E=84.4 \text{ tn/mm}^2$  and limit elastic strength  $f=0.2283 \text{ tn/mm}^2$ , and plastic flow along the fibre direction has been assumed in the fictitious isotropic space.

Perfect plastic behaviour for both materials has been assumed.

Figure 10(1) displays the stress-strain behaviour for the composite with longitudinal fibres, showing the behaviour of the composite (curve b), the matrix (curve c) and the fibres (curve a).

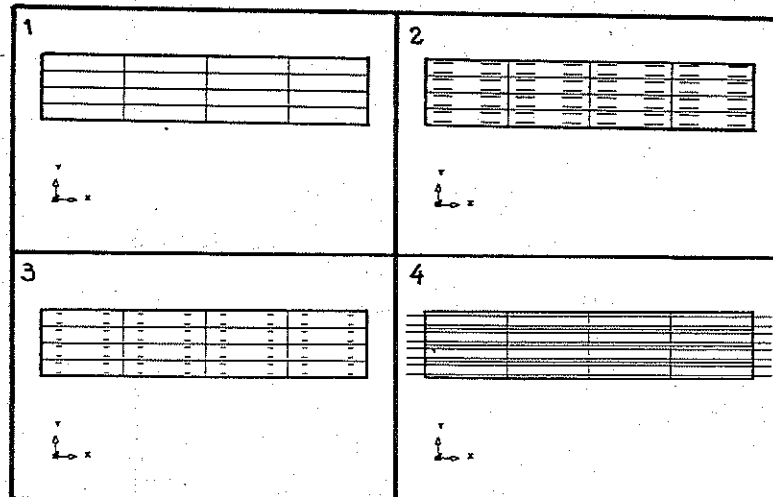


Figure 12 Principal stresses for the longitudinal fibre case. 1, Finite element mesh; 2, principal stresses in the composite; 3, principal stresses in the matrix; 4, principal stresses in the fibres

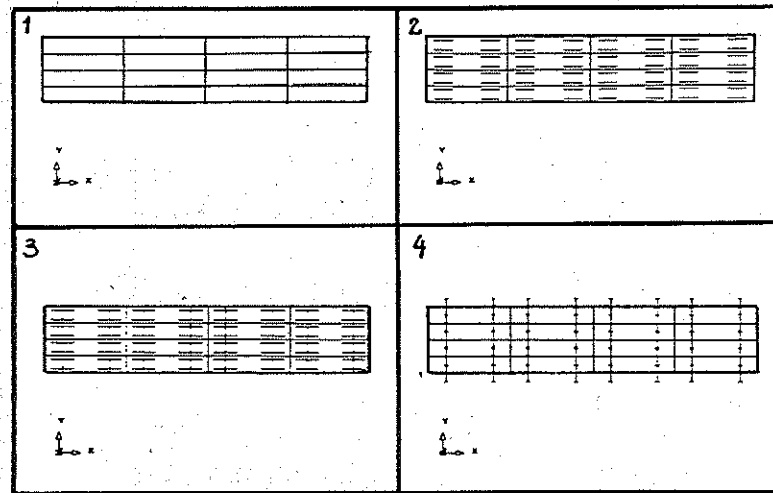


Figure 13 Principal stresses for the transverse fibre case. 1, Finite element mesh; 2, principal stresses in the composite; 3, principal stresses in the matrix; 4, principal stresses in the fibres

Also, the same Figure shows that the stiffness of the composite remain equal to that of the matrix after plastification of the fibres.

Figure 10(2) shows the stress-strain behaviour for the composite with transversal fibres. Curve (a) shows the evolution of the longitudinal stress in the composite, curve (b) shows that the strength of the matrix is greater than that of the composite. This can be explained by the small contribution of the transverse fibres to the global longitudinal stiffness. Therefore, the resisting material has 20% less transverse cross area with the corresponding reduction in stiffness. Curve (c) in the same Figure shows the transversal tension stress in the matrix and the corresponding



compression stress in the fibres, curve (d). Finally curve (e) shows the null 'global' resistance of the composite in the transverse direction as expected.

Figure 11 shows the load-displacement curves for the two fibre orientations studied.

Figures 12 and 13 show, in qualitative form, the principal stress field in the composite material, as well as in the matrix and the fibre components, for each of two fibre orientation case analyses, respectively.

The numerical results obtained are in good agreement with the expected values, thus showing the ability of the model proposed to analyse fibre-reinforced composites.

### CONCLUDING REMARKS

It has been shown that the elasto-plastic anisotropic behaviour of solids can be effectively modelled by means of an equivalent isotropic model based on standard Green-Naghdi large strain elasto-plasticity theory. The model proposed is particularly suited for finite element computations of composite structures with different material properties. Initial applications of the model to standard isotropic materials and more complex fibre-reinforced composites using a multiphase material model<sup>25</sup> show promising results which will be more extensively validated in the near future. Extensions of the anisotropic formulation presented to non-linear damage mechanics are straightforward and are currently investigated by the authors.

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